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Hu, Wenhui; Yang, Fan; Pietraszak, Nick; Gu, Jing; and Huang, Jier, "Distance Dependent Energy Transfer Dynamics from a Molecular Donor to a Zeolitic Imidazolate Framework Acceptor" (2020). *Chemistry Faculty Research and Publications*. 1031.

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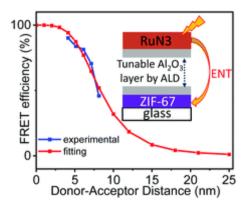
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# Distance Dependent Energy Transfer Dynamics from a Molecular Donor to a Zeolitic Imidazolate Framework Acceptor

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## Abstract

Zeolitic Imidazolate frameworks (ZIFs) have been demonstrated as promising light harvesting and photocatalytic materials for solar energy conversion. To facilitate their application in photocatalysis, it is essential to develop a fundamental understanding of their light absorption properties and energy transfer dynamics. In this work, we report distance-dependent energy transfer dynamics from a molecular photosensitizer (RuN3) to ZIF-67, where the distance between RuN3 and ZIF-67 is finely tuned by depositing an ultrathin Al<sub>2</sub>O<sub>3</sub> layer on the ZIF-67 surface using an atomic layer deposition (ALD) method. We show that energy transfer time decreases with increasing distance between RuN3 and ZIF-67 and the Förster radius is estimated to be 14.4 nm.



# Introduction

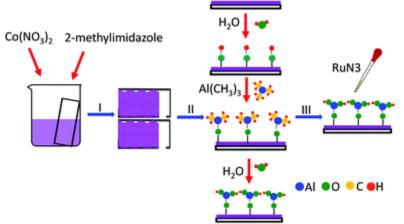
Zeolitic imidazolate frameworks (ZIFs), a subclass of metal organic frameworks (MOFs), are composed of Zn<sup>2+</sup> or Co<sup>2+</sup> nodes tetrahedrally coordinated with imidazole-based organic linkers.<sup>1–5</sup> Owing to their ordered porous structure and large surface area, ZIFs have emerged as new materials for gas storage and separation,<sup>6–9</sup> chemical sensing,<sup>10,11</sup> and catalysis.<sup>12–14</sup> Driven by the demand for renewable energy and environmental concerns, recent interests have extended their application in photocatalysis with a number of reports having demonstrated their capability as photocatalytic materials.<sup>15–19</sup> However, in the majority of these systems, ZIFs were either used as hosts for reaction substrates/catalytic active species or templates to synthesize porous hybrid materials through an annealing process.<sup>20–24</sup> In contrast, our recent studies showed that ZIFs based on Co nodes and 2-methyl imidazolate ligand (ZIF-67) not only possess broad absorption in the UV-visible-near IR region but also exhibit a long-lived ES.<sup>25,26</sup> A further study then showed that the electron in this ES state can be extracted through interfacial electron transfer (ET) from excited ZIF-67 to methylene blue, which largely demonstrates the promise of using ZIFs as intrinsic light harvesting and charge separation materials for solar energy conversion.<sup>27</sup>

While ZIF-67 has broad absorption in both the visible and near IR region, the extinction coefficients of these spectral transitions resulting from dipole forbidden d–d transitions of Co nodes are quite low (~100–1000 mol  $L^{-1}$  cm<sup>-1</sup>).<sup>28</sup> In response to this challenge, we encapsulated molecular (RuN3)<sup>29</sup> and semiconductor (CdS)<sup>30</sup> photosensitizers (PS), which have absorption in the visible region that compensates the absorption of ZIF-67 and have much larger extinction coefficient, into ZIF-67. We showed that both systems can strengthen the light harvesting ability of ZIF-67 as efficient energy transfer (ENT) can occur from the guest PSs to ZIF-67. These results demonstrate that encapsulating a guest unit chromophore that can relay energy to ZIFs through ENT is a promising approach to enhance the light harvesting ability of ZIFs. A natural question that follows these ENT studies is to unravel the key factors that control the dynamics of ENT. It has been shown previously that ENT efficiency is largely dependent on the distance between the donor and acceptor.<sup>31–38</sup> In this work, we report the impact of distance between RuN3 and ZIF-67 on the ENT dynamics in the RuN3/ZIF-67 hybrid. The distance

between RuN3 and ZIF-67 is controlled by tuning the thickness of the  $Al_2O_3$  layer from 3 nm to 8.5 nm, which is deposited on the surface of the ZIF-67 film before sensitization of RuN3 using atomic layer deposition (ALD). We show that the ENT efficiency decreases with increasing thickness of  $Al_2O_3$  between RuN3 and ZIF-67, where the theoretical Förster radius estimated according to the reported point to plane resonance energy transfer under 4th-power law<sup>33,39,40</sup> is 14.4 nm.

#### Results and discussion

The schematic representation of the synthesis of RuN3/Al<sub>2</sub>O<sub>3</sub>/ZIF-67 hybrid films is illustrated in Scheme 1 (see details in ESI<sup>+</sup>). In the first step, a glass slide pre-treated with Piranha solution was immersed into the mixture of Co(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O and 2-methylimidazole (2 mIm). After about 1 h, transparent and continuous ZIF-67 film was formed on both sides of the glass slide (step I).<sup>25,29</sup> ZIF-67 crystals on one side of the film are scratched off to make a single-side ZIF-67 film. The Al<sub>2</sub>O<sub>3</sub> layer with different thickness was then deposited on the surface of ZIF-67 film using ALD (step II). At a temperature of 100–200 °C, the deposition of Al<sub>2</sub>O<sub>3</sub> usually has a stable growth rate of 0.1–0.11 nm per cycle on non-porous substrates.<sup>41–43</sup> However, as demonstrated using the transmission electron microscopy (TEM) images, a much thicker layer of Al<sub>2</sub>O<sub>3</sub> was identified (Fig. S1, ESI<sup>+</sup>). This can be explained by the nanoporous structure of ZIF-67, where Al<sub>2</sub>O<sub>3</sub> is not only deposited on the surface but also the subsurface in the nanostructure. As a result, alternatively depositing 10 to 40 cycles of trimethylaluminum and water at 120 °C resulted in a 3 nm to 8.5 nm Al<sub>2</sub>O<sub>3</sub> thin film on ZIF-67 (Table S1, ESI<sup>+</sup>). After the deposition of Al<sub>2</sub>O<sub>3</sub>, the same amount of RuN3 in methanol solution was dropped onto the Al<sub>2</sub>O<sub>3</sub>/ZIF-67 films to form RuN3/Al<sub>2</sub>O<sub>3</sub>/ZIF-67 hybrid films (step III).



Scheme 1 Schematic representation of the synthesis of RuN3/Al<sub>2</sub>O<sub>3</sub>/ZIF-67 thin film.

Fig. 1a shows the XRD patterns of blank glass slides,  $Al_2O_3$  on glass slides, ZIF-67 on glass slides,  $Al_2O_3(8.5 \text{ nm})/\text{ZIF-67}$ , and  $RuN3/Al_2O_3(8.5 \text{ nm})/\text{ZIF-67}$ . A broad peak was observed in the range of 20° to 40° among all samples including the naked glass slide, which can be attributed to the diffraction of amorphous glass.  $Al_2O_3/\text{ZIF-67}$  and  $RuN3/Al_2O_3/\text{ZIF-67}$  films (Fig. 1a and Fig. S2a and b) all show similar XRD patterns to ZIF-67 film on glass, suggesting that the ZIF-67 structure is retained in the films after ALD deposition of  $Al_2O_3$  and RuN3 sensitization. Note that the diffraction patterns of the ZIF-67 film on glass slide on the diffraction patterns as the XRD patterns of the ZIF-67 crystals are scratched off from the glass slide resemble that of ZIF-67 crystals synthesized from standard growth (Fig. S2c, ESI<sup>+</sup>). The retention of the ZIF-67 structure in these hybrid films was further supported by the UV-Visible absorption spectra (Fig. 1b and Fig. S3, ESI<sup>+</sup>), where  $Al_2O_3/\text{ZIF-67}$  and  $RuN3/Al_2O_3/\text{ZIF-67}$  films with different thickness of  $Al_2O_3$  all show absorption peaks centered at 585 nm originating from  $T_d$  CO<sup>II</sup> d–d transition, consistent with that of ZIF-67.<sup>25</sup> While RuN3 has a prominent absorption

peak around 500 nm corresponding to ligand-to-metal charge transfer band (LMCT) (pink plot in Fig. 1b), it cannot be easily seen from the UV-visible absorption spectrum of RuN3/ZIF-67 due to its overlap with ZIF-67 absorption. Nevertheless, the transient absorption experiments below confirm the adsorption of RuN3 on the surface of  $Al_2O_3/ZIF-67$  film.

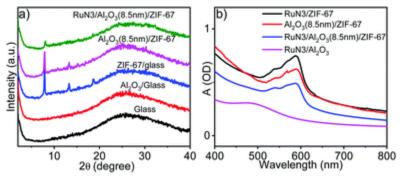


Fig. 1 (a) XRD patterns of glass, Al<sub>2</sub>O<sub>3</sub>/glass, ZIF-67/glass, Al<sub>2</sub>O<sub>3</sub>(8.5 nm)/ZIF-67 and RuN3/Al<sub>2</sub>O<sub>3</sub>(8.5 nm)/ZIF-67. (b) UV-visible absorption spectra of RuN3/ZIF-67, Al<sub>2</sub>O<sub>3</sub>(8.5 nm)/ZIF-67, RuN3/Al<sub>2</sub>O<sub>3</sub>(8.5 nm)/ZIF-67 and RuN3/Al<sub>2</sub>O<sub>3</sub>.

Transient absorption (TA) spectroscopy is used to examine the impact of the thickness of the  $Al_2O_3$  layer on the ENT dynamics from RuN3 to ZIF-67. Fig. 2a shows the TA spectra of RuN3/ZIF-67 following 410 nm excitation, which selectively excites RuN3 as ZIF-67 has negligible absorption at 410 nm. Consistent with the previous literature result,<sup>29</sup> immediately following the excitation, the TA spectra of RuN3/ZIF-67 show a negative ground state bleach (GSB) centered at ~530 nm and a broad positive excited state absorption (ESA) feature at >570 nm, which is due to the excitation of RuN3, resulting in the depopulation of the RuN3 ground state and population of the RuN3 excited state. As the lifetime of excited singlet state of LMCT (<sup>1</sup>LMCT) is <100 fs,<sup>44,45</sup> which is much faster than our instrument response time ( $\sim$ 200 fs), the ESA absorption of RuN3 can be attributed to <sup>3</sup>LMCT. The GSB of RuN3 recovers and ESA decays with time simultaneously (Fig. 2b), which is accompanied by the formation of a derivative feature consisting of a negative feature centered at 585 nm and absorption at 605 nm at later time (>200 ps), consistent with the typical spectral features of the excited state of ZIF-67 corresponding to <sup>4</sup>A<sub>2</sub>(F)–<sup>4</sup>T<sub>1</sub>(P) Co d–d transition,<sup>17,25</sup> suggesting that the excitation of RuN3 leads to the formation of excited ZIF-67.<sup>29</sup> Moreover, the GSB recovery and ESA decay in RuN3/ZIF-67 are much faster than that of RuN3/Al<sub>2</sub>O<sub>3</sub> (Fig. 2b), where the latter is used as a model system for intrinsic ES dynamics of RuN3 on a solid surface as ENT from RuN3 to  $Al_2O_3$  is not expected due to significantly larger band gap of  $Al_2O_3$  than RuN3.<sup>46–</sup> <sup>48</sup> These results together support that ENT occurs from RuN3 (<sup>3</sup>LMCT) to ZIF-67 (<sup>4</sup>A<sub>2</sub>) following the excitation of RuN3, which guenches the ES of RuN3 and results in the formation of ZIF-67 ES, consistent with the previous report.<sup>29</sup> With the presence of an  $Al_2O_3$  layer (3 nm thickness) between RuN3 and ZIF-67 (Fig. 2c), the TA spectra of RuN3/Al<sub>2</sub>O<sub>3</sub>(3 nm)/ZIF-67 resemble that of RuN3/ZIF-67. However, with increasing thickness of Al<sub>2</sub>O<sub>3</sub>, the derivative feature corresponding to ES of ZIF-67 in RuN3/Al<sub>2</sub>O<sub>3</sub>/ZIF-67 becomes weaker and weaker (Fig. 2d, e and Fig. S4, ESI<sup>+</sup>) and can be barely seen when the thickness of Al<sub>2</sub>O<sub>3</sub> is 8.5 nm (Fig. 2e). These results suggest that the ENT process is partially blocked by  $Al_2O_3$  due to its inert nature, which results in decreasing ENT rate with increasing thickness of Al<sub>2</sub>O<sub>3</sub>.

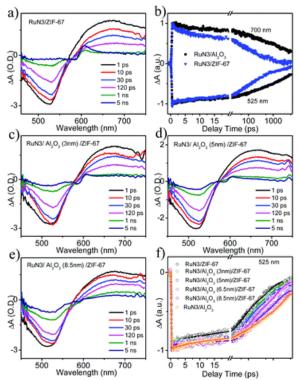


Fig. 2 Transient absorption spectra of RuN3/ZIF-67 (a), RuN3/Al<sub>2</sub>O<sub>3</sub>(3 nm)/ZIF-67 (c), RuN3/Al<sub>2</sub>O<sub>3</sub>(5 nm)/ZIF-67 (d), and RuN3/Al<sub>2</sub>O<sub>3</sub>(8.5 nm)/ZIF-67 (e). (b) The comparison of GSB recovery and ESA decay kinetics of RuN3 on ZIF-67 and Al<sub>2</sub>O<sub>3</sub>. (f) The comparison of GSB recovery kinetics of RuN3 on different substrates.

The dependence of the ENT process on Al<sub>2</sub>O<sub>3</sub> thickness can be more clearly seen from the comparison of the GSB kinetics of RuN3 at 525 nm (Fig. 2f) or ESA of RuN3 (Fig. S5, ESI<sup>+</sup>) among RuN3/Al<sub>2</sub>O<sub>3</sub>/ZIF-67 samples with different thickness of Al<sub>2</sub>O<sub>3</sub>. As shown in Fig. 2f, the GSB recovery kinetics of RuN3/Al<sub>2</sub>O<sub>3</sub>/ZIF-67 become slower with increasing thickness of Al<sub>2</sub>O<sub>3</sub>, consistent with the assignment above. Since the recovery lifetime of GSB of these RuN3/Al<sub>2</sub>O<sub>3</sub>/ZIF-67 samples is much longer than 5 ns, which is beyond our TA time window, the ENT time was calculated based on the half lifetime ( $\tau_{1/2}$ ), which is the time that the kinetic trace decays to half of its maximum amplitude. As listed in Table 1,  $\tau_{1/2}$  for RuN3/Al<sub>2</sub>O<sub>3</sub>/ZIF-67 with 0 nm, 3 nm, 5 nm, 6.5 nm, and 8.5 nm is 96 ps, 158 ps, 182 ps, 287 ps, and 528 ps, respectively. According to these half lifetimes, we estimated the ENT time according to eqn (1).1/ $\tau_{1/2} = 1/\tau_0 + 1/\tau_{ENT}(1)\eta = \tau_{1/2}/\tau_{ENT}(2)$ where  $\tau_{ENT}$  is the ENT time from RuN3 to ZIF-67 and  $\tau_0$  is the intrinsic ES decay time of RuN3. ENT efficiency ( $\eta$ ) can then be calculated according to eqn (2). The calculated ENT efficiency is also listed in Table 1. The ENT efficiency decreased almost half (from 90.6% to 46.0%) when the thickness of the Al<sub>2</sub>O<sub>3</sub> thin film increased to 8.5 nm, indicating that the ENT efficiency in the RuN3/ZIF-67 system is sensitive to the distance between RuN3 and ZIF-67.

**Table 1** The half lifetime of RuN3 GSB on different films and estimated ENT time and efficiency ofRuN3/Al $_2O_3$ /ZIF-67 films

	τ <sub>1/2</sub> (ps)	τ <sub>ENT</sub> (ps)	$\eta_{\rm ENT}$ (100%)
RuN3/ZIF-67	96	106	90.6
RuN3/Al <sub>2</sub> O <sub>3</sub> (3 nm)/ZIF-67	158	188	84.0
RuN3/Al <sub>2</sub> O <sub>3</sub> (5 nm)/ZIF-67	182	223	81.6
RuN3/Al <sub>2</sub> O <sub>3</sub> (6.5 nm)/ZIF-67	287	406	70.7
RuN3/Al <sub>2</sub> O <sub>3</sub> (8.5 nm)/ZIF-67	528	1147	46.0
RuN3/Al <sub>2</sub> O <sub>3</sub> ( $\tau_0$ )	978	—	—

The theoretical Förster radius of this system was estimated by fitting the experimental data using eqn (3):<sup>33,39,40</sup> $\eta = 1/[1 + (R/R_0)^4](3)$  where  $R_0$  and R are the Förster radius and distance between the donor and acceptor, respectively.  $R_0$  equals R when the ENT efficiency reaches 50%. In the fitting process, the distance between ZIF-67 and RuN3 without Al<sub>2</sub>O<sub>3</sub> (r) and the Förster radius ( $R_0$ ) were used as fitting parameters, where R is the sum of r and the thickness of the Al<sub>2</sub>O<sub>3</sub> layer. As shown in Fig. 3, the experimental results can be adequately fit by the proposed model. From the best fitting, we obtained an r value of 5.7 nm and  $R_0$  of 14.4 nm. The  $R_0$  value in this system is much higher than the previously reported molecular donor/acceptor system (<5 nm).<sup>29,49,50</sup> Given that a larger  $R_0$  value results in higher FRET efficiency (eqn (3)),<sup>50,51</sup> the much higher value in the current RuN3/ZIF-67 than the molecular systems suggests that the framework of ZIF might be beneficial for the ENT process. In addition, a larger  $R_0$  value can typically facilitate long–range energy transfer,<sup>52</sup> which suggests the potential of further enhancing the light absorption ability of ZIF systems through controlling the ENT process.

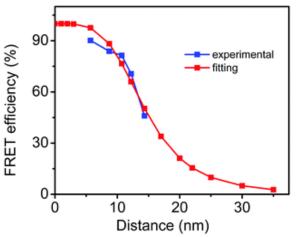


Fig. 3 The ENT efficiency as a function of distance between ZIF-67 and RuN3.

#### Conclusion

In summary, we report the ENT dynamics from RuN3 to ZIF-67 and the impact of distance between RuN3 (donor) and ZIF-67 (acceptor) on ENT efficiency, where the distance between the donor and acceptor was controlled by tuning the thickness of the Al<sub>2</sub>O<sub>3</sub> film deposited on the surface of ZIF-67 using atomic layer deposition (ALD). Using transient absorption spectroscopy, we show that the ENT efficiency decreases with increasing thickness of the Al<sub>2</sub>O<sub>3</sub> layer between RuN3 and ZIF-67. According to these experimental results, the Förster radius for this system was estimated to be 14.4 nm, which is much larger than many molecular donor/acceptor systems, suggesting the promise of enhancing the light harvesting capability of ZIFs through the ENT process. This work not only demonstrates the capability to tuning the distance of the donor and acceptor by depositing different-thickness Al<sub>2</sub>O<sub>3</sub> layers using ALD but also provides new insight into controlling the ENT dynamics in RuN3/ZIF-67.

# Author contributions

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

# Conflicts of interest

There are no conflicts to declare.

#### Acknowledgements

This work was supported by National Science Foundation (DMR-1654140) and ACS-PRF (57503-DNI6). Wenhui Hu acknowledge the John J. Eisch fellowship during the 2019–2020 academic year.

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